

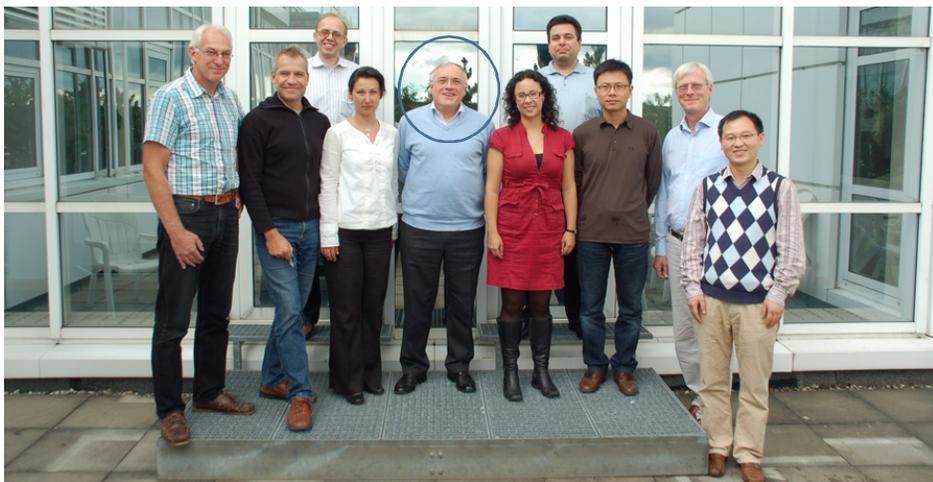
Career Review

(Klaus Mullen)

06/10/2015

Ki-Young Yoon

Prof. Klaus Müllen

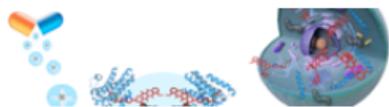


at Max Planck Institute
for Polymer Research, Germany

- Born in 1947
- Ph.D. in 1971 at University of Basel
- Post Doc. at ETH Zurich
- Independent career (1979-Present)
- Director at MPIP (1989-Present)
- JACS associate editor (2006-Present)

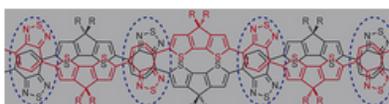
>1700 papers, H index=117

Research Interests



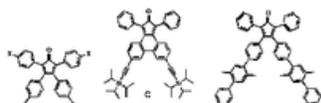
Biomaterials

Dr. Kalina Peneva



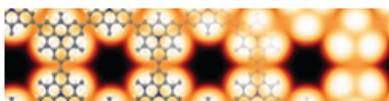
Conjugated Polymers

Prof. Dr. Martin Baumgarten



Dendritic Materials

Prof. Dr. Martin Baumgarten



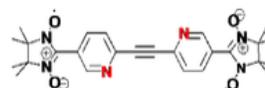
Graphite Group

Prof. Dr. Xinliang Feng / Dr. Manfred Wagner



High Performance Rylene Dyes

Dr. Long Chen



High Spin Organics

Prof. Dr. Martin Baumgarten



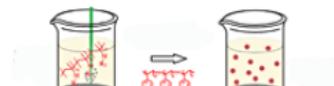
Mass Spectrometry

Dr. Hans Joachim Räder



NMR Spectroscopy

Dr. Manfred Wagner



Polymerisation Procedures

Dr. Markus Klapper



Material Science

Dr. Wojciech Pisula



Inorganic-Organic Hybrid Systems

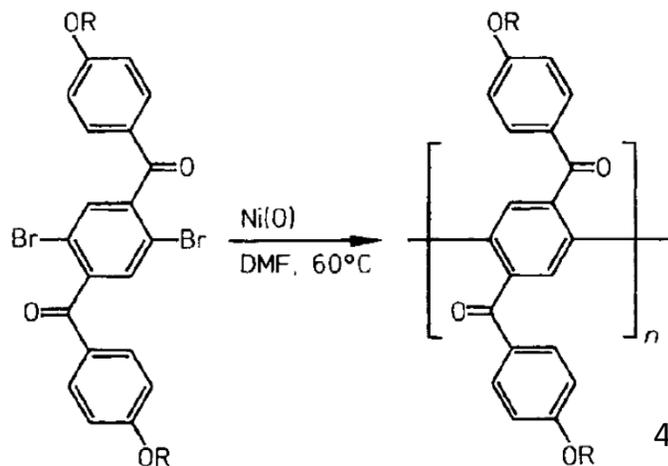
PD Dr. Hermann Sachdev

Mullen's webpage

<http://www.mpip-mainz.mpg.de/17142/Research>

Graphene nanoribbon

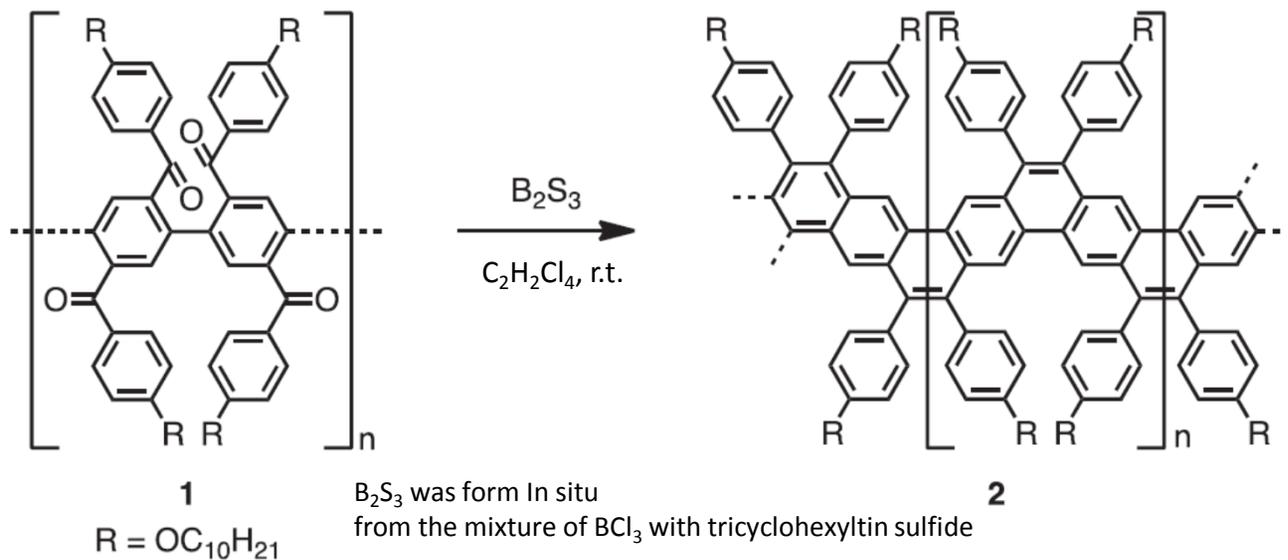
In 1993



Yamamoto polymerization,
followed by carbonyl olefination

4000 Da (PDI=1.37)

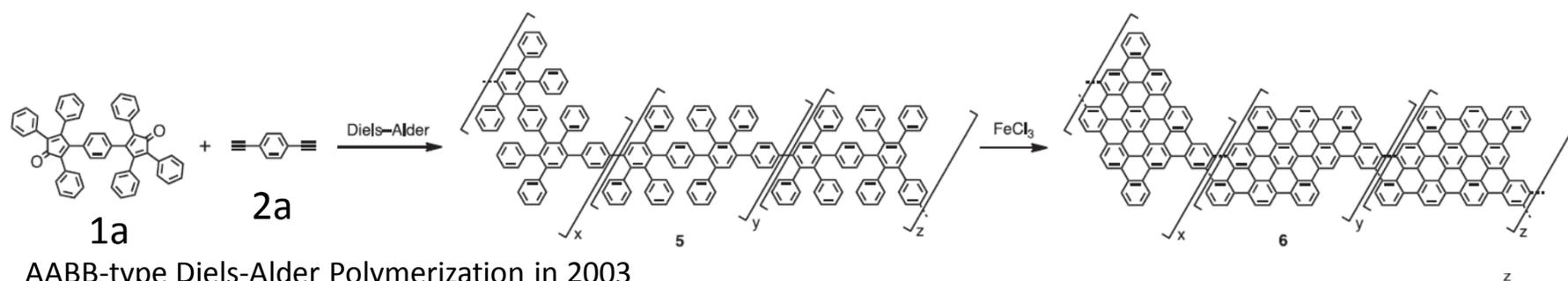
They hadn't realized
this was a graphene nanoribbon



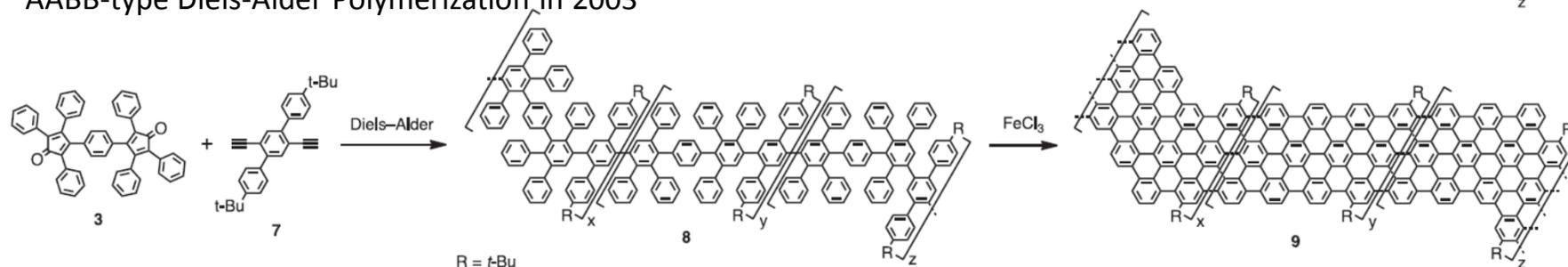
Graphene Nanoribbons (Solution Synthesis)

AABB-type Diels-Alder Polymerization in 2000

“Graphite ribbon”



AABB-type Diels-Alder Polymerization in 2003



$M_n = 6200$ (PDI=1.9) – $M_n = 30100$ (PDI=4.0)

Table 1. Reaction Conditions and Characterization of the Polymers (3a) Prepared by the Diels–Alder Polycondensation between 1,4-Bis(2,4,5-triphenylcyclopentadienone-3-yl)benzene (1a) and 1,4-Diethynylbenzene (2a)

runs	concn of 1a and 2a [mol/L]	time of reaction [h]	yield [%]	$M_w \times 10^{-3}{}^a$ [g mol ⁻¹] (SEC)	$M_n \times 10^{-3}{}^a$ [g mol ⁻¹] (SEC)	M_w/M_n	MALDI-TOF (detected species)
a	0.3	18	92	12.3	6.2	1.9	16mer
b	0.3	30	94	17.4	8.6	2.0	18mer
c	0.5	18	91	32.2	14.1	2.3	18mer
d	0.5	30	92	40.6	16.8	2.5	<i>b</i>
e	0.5	48	94	51.5	19.7	2.7	<i>b</i>
f	0.7	48	93	101.4	26.6	3.9	<i>b</i>
g	0.7	72	94	121.6	30.1	4.0	<i>b</i>

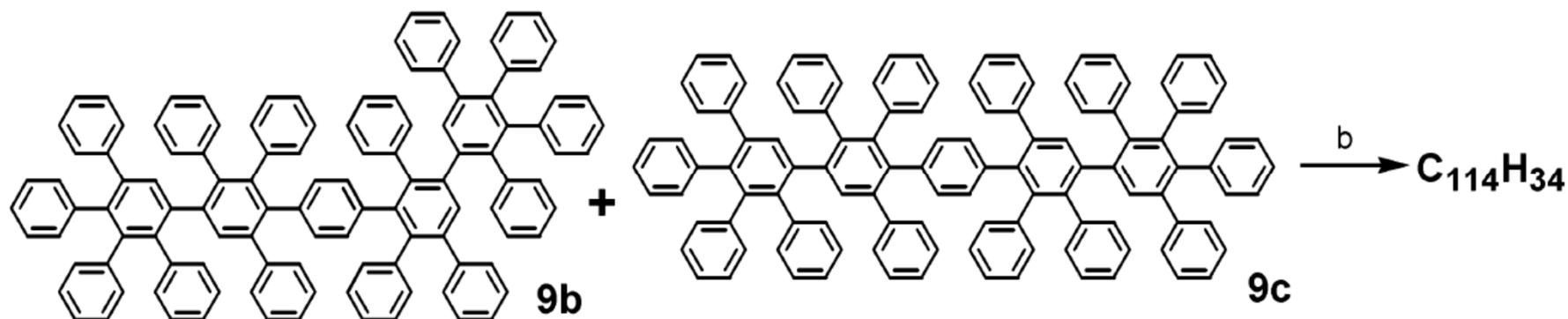
^a Determined with polystyrene as standard. ^b Measurements were not performed.

Higher monomer concentration
Longer reaction time
=> Higher MW

Macromolecules **2000**, *33*, 3525

Macromolecules **2003**, *36*, 7082

How to characterize the polymers after Scholl oxidation



(a) Ph_2O , reflux; (b) FeCl_3 , CH_2Cl_2 , CH_3NO_2 , 20 h.

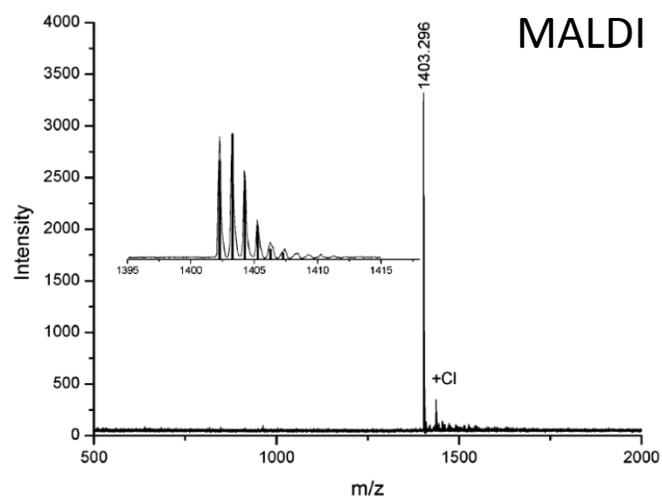


Figure 1. MALDI MS spectrum for $\text{C}_{114}\text{H}_{34}$: calculated, 1403.54; found, 1403.30 (100%). Isotope distribution is in good agreement with the simulated results (black bar). In addition, some chlorination took place during the cyclodehydrogenation with Lewis acid iron(III) chloride.

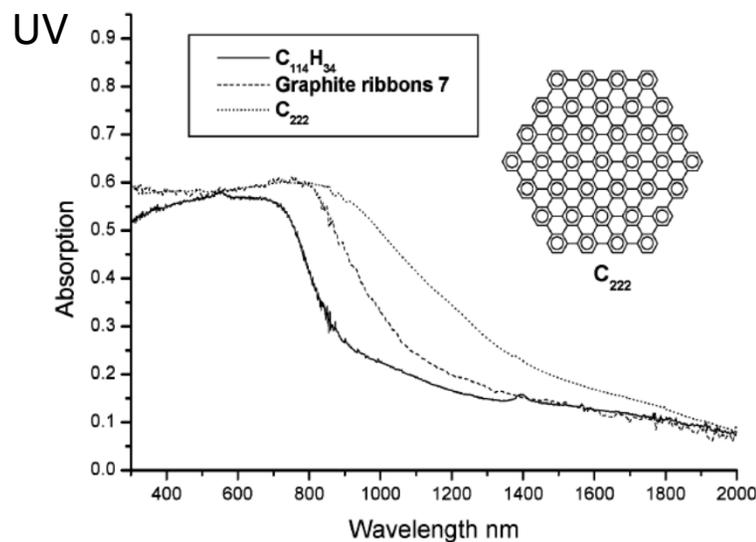


Figure 2. Solid-state UV-vis spectrum of graphite ribbon 7, $\text{C}_{114}\text{H}_{34}$, and C_{222} .

How to characterize the polymers after Scholl oxidation

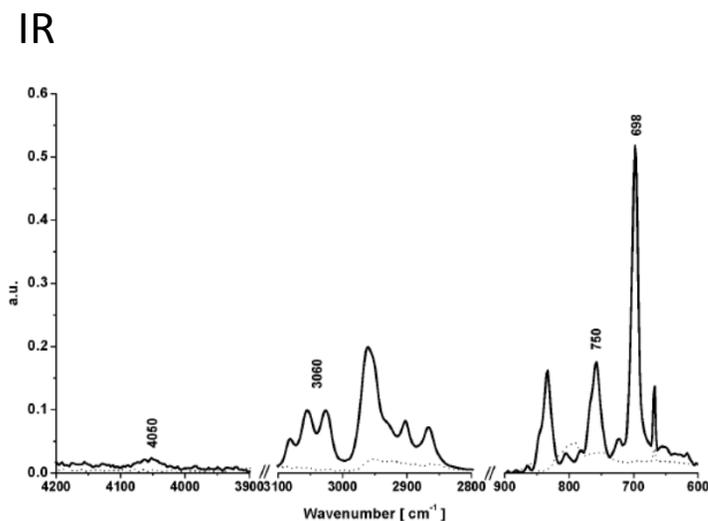


Figure 3. Infrared spectra of branched polyphenylene precursor **6** (solid line) and graphite ribbon **7** (dot line).

Disappearance of
monosubstituted benzene peaks
& “free” rotated benzene peak

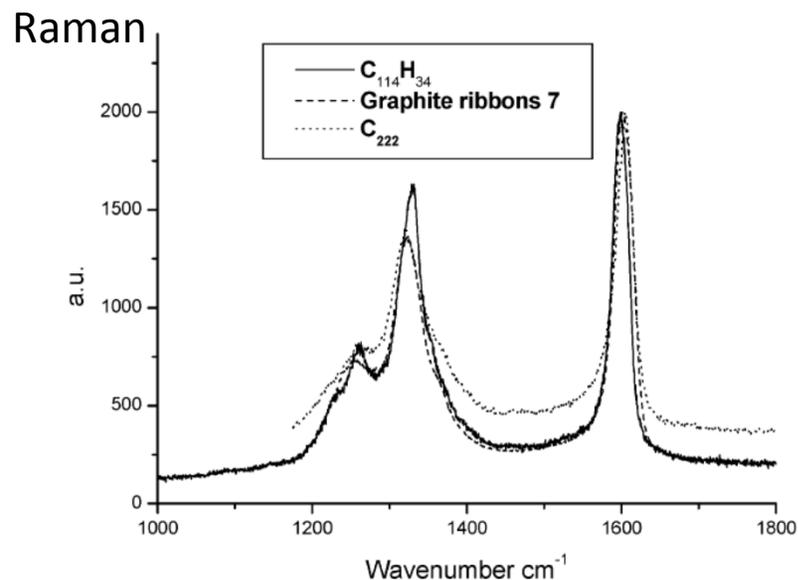


Figure 4. Raman spectra of graphite ribbon **7**, **C₁₁₄H₃₄**, and **C₂₂₂**.

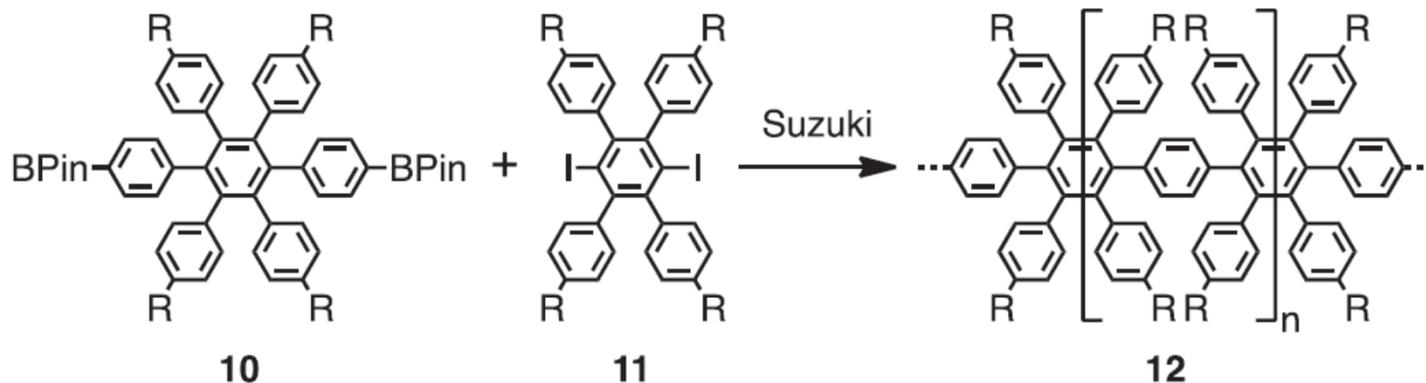
Existence of
G band (~1600 cm⁻¹) & D band (~1300 cm⁻¹)

+ Microscopic analyses

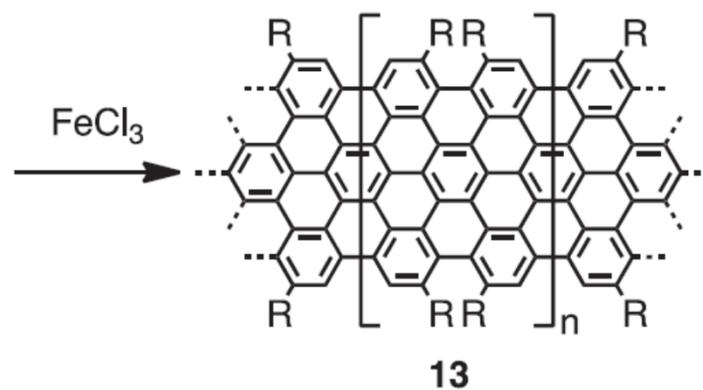
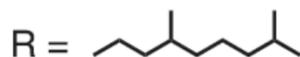
Graphene Nanoribbons (Solution Synthesis)

AABB-type Suzuki Polymerization in 2008

“Graphene Nanoribbon”



$M_n=14000$ (PDI=1.2) after soxhlet extraction

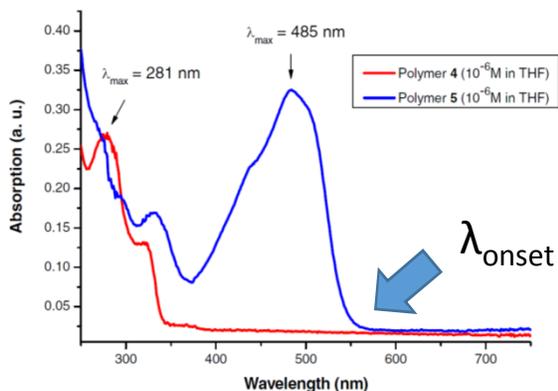


N=9 GNR

Only up to 12 nm GNR

Graphene Nanoribbons (Solution Synthesis)

AABB-type Suzuki Polymerization in 2008



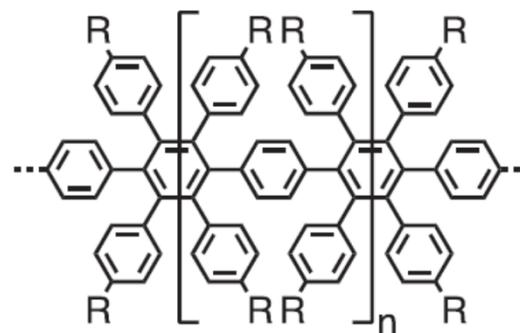
$$\lambda_{\text{onset}} \approx 550 \text{ nm}$$

2.3 eV
Too large bandgap
(if infinite length, 1 eV)

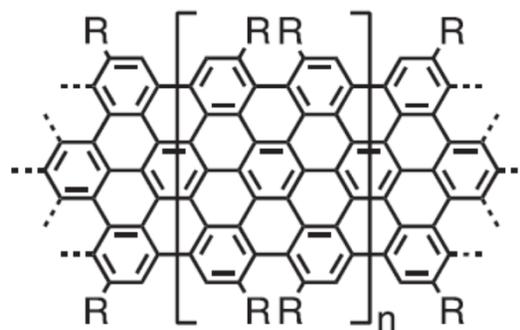
Tip : $1240 / \lambda_{\text{onset}} = \text{Bandgap (eV)}$

The low efficiency of polymerization was presumably

- i) High steric hindrance
- ii) Limited solubility due to rigid PPP backbone



12



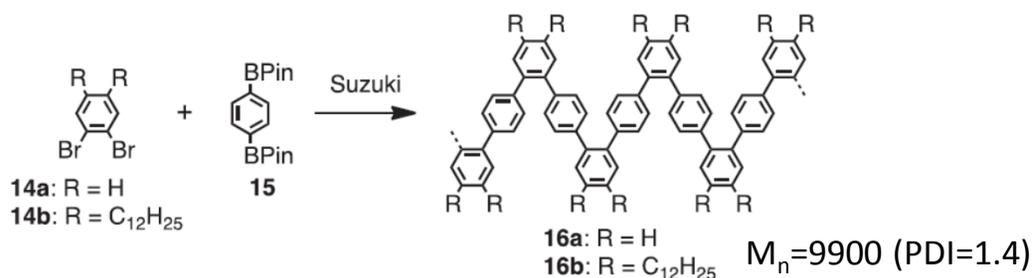
13

N=9 GNR

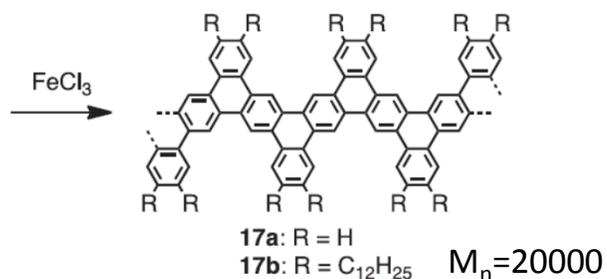
Only up to 12 nm GNR

Graphene Nanoribbons (Solution Synthesis)

AABB-type Suzuki Polymerization in 2011



Less steric hindrance,
Better solubility (Flexible polymer)
- 40 mg/mL in DCM, THF, Toluene



No bandgap information

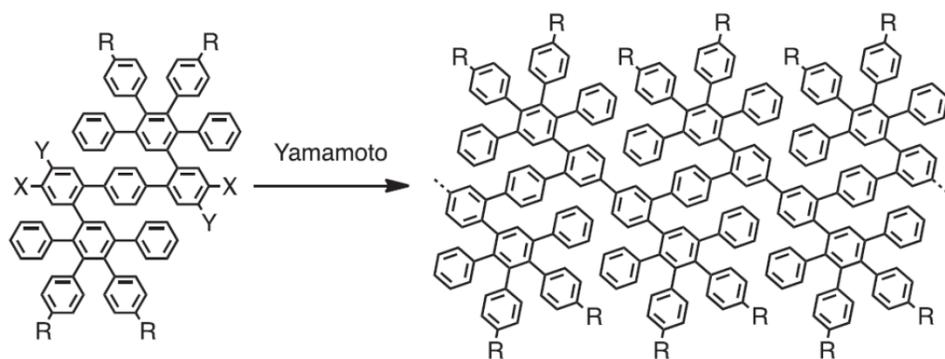
25 nm for GNR
(no evidence)

Q3

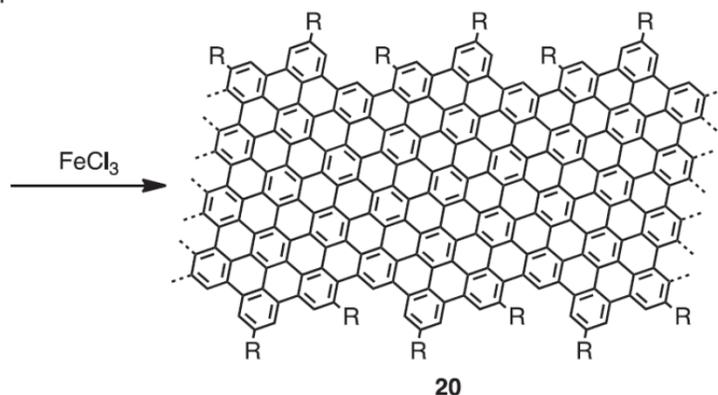
A paper that the main script doesn't match with SI.

Graphene Nanoribbons (Solution Synthesis)

AA-type Yamamoto Polymerization in 2012



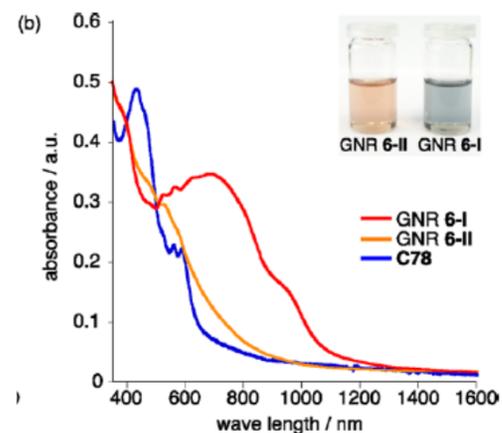
18a: X = H, Y = Cl
18b: X = C₁₂H₂₅, Y = H
R = C₁₂H₂₅



No problem of stoichiometric imbalance
, which AABB-type polymerization always has.

From polymer with $M_n=13000$ (PDI=2.2),
(by prep GPC)

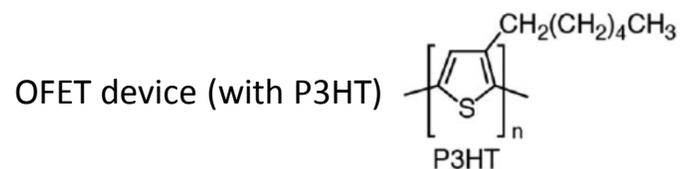
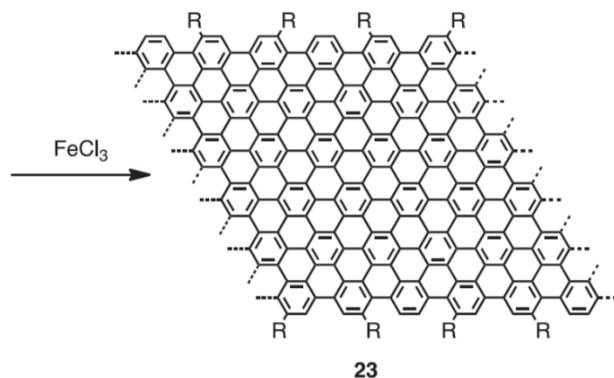
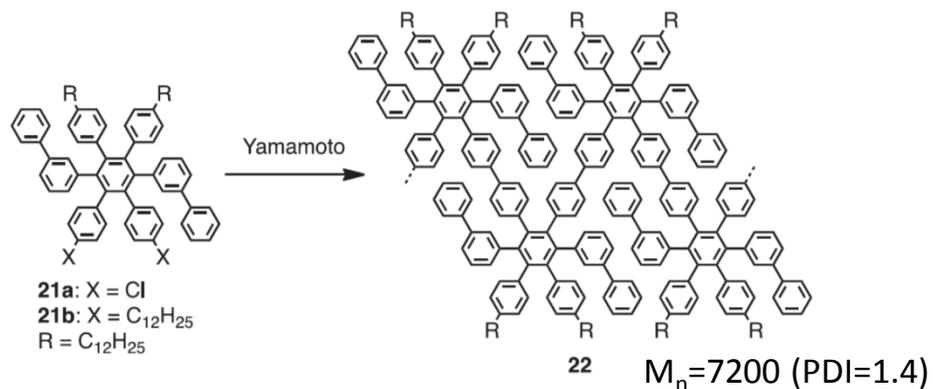
High MW part, $M_n=44000$ (PDI=1.2)
Low MW part, $M_n=6700$ (PDI=1.1)



20-30 nm length based on M_n & Bandgap
(No microscopic data)

Graphene Nanoribbons (Solution Synthesis)

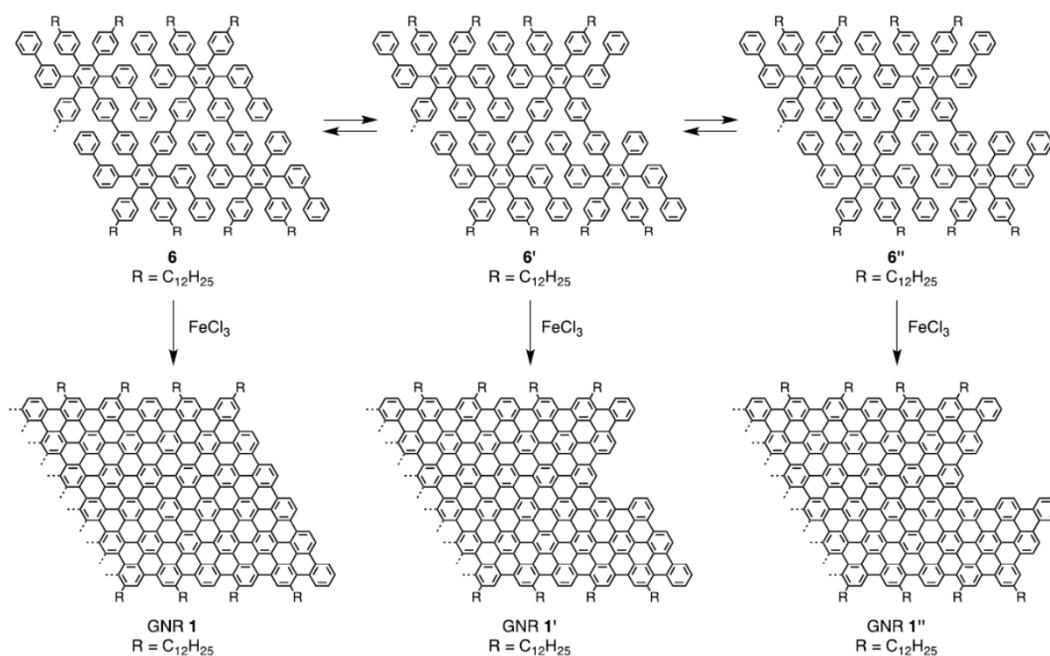
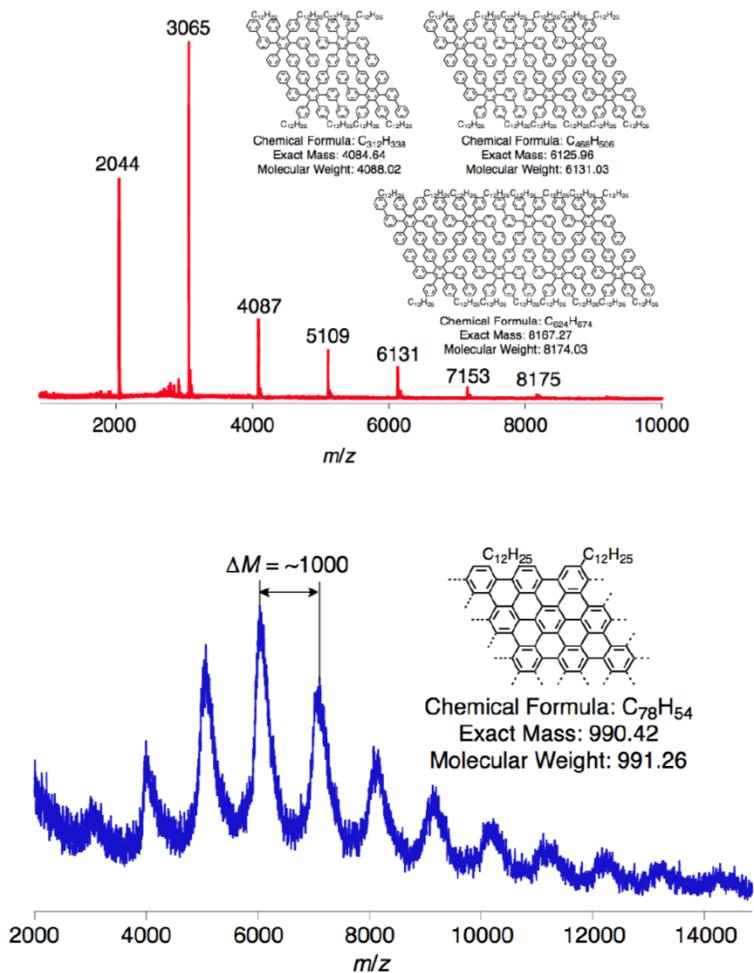
AA-type Yamamoto Polymerization in 2014



Optical bandgap= \sim 1.6 eV \Rightarrow **Q1**

Graphene Nanoribbons (Solution Synthesis)

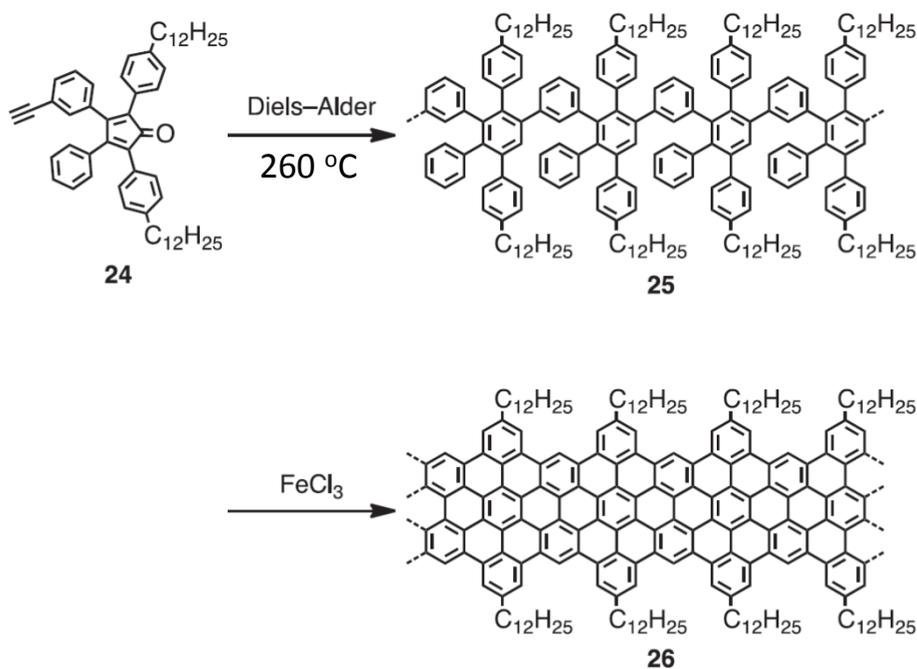
AA-type Yamamoto Polymerization in 2014



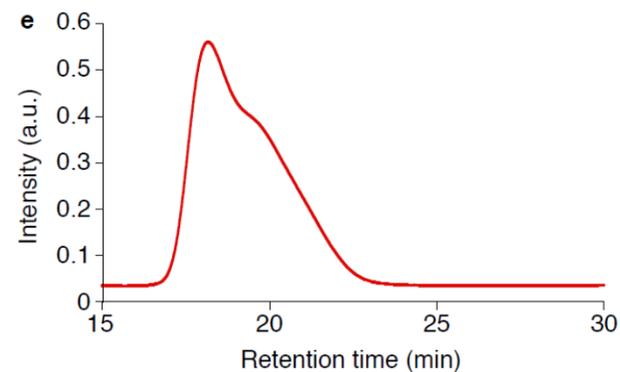
Very broad mass spec

Graphene Nanoribbons (Solution Synthesis)

AB-type Diels-Alder Polymerization in 2014

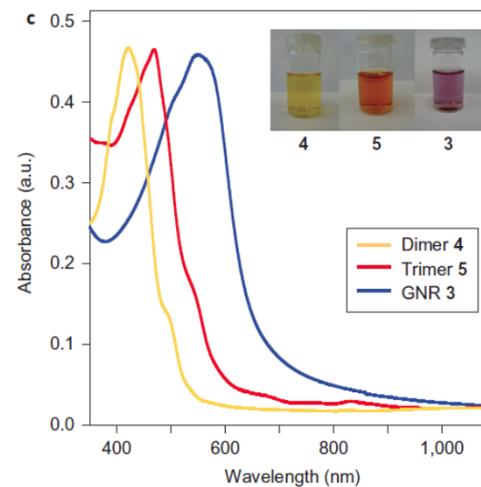


GPC trace



$M_n=340,000$ (PDI=1.9) after fractionation.

UV/vis



Bandgap
=1.88 eV

Graphene Nanoribbons (Solution Synthesis)

AB-type Diels-Alder Polymerization in 2014

Entry	Solvent	<i>C</i> (mM)	<i>T</i> (h)	$M_{w,PPP}-M_{w,PS}$ (kgmol ⁻¹)	$M_{n,PPP}-M_{n,PS}$ (kgmol ⁻¹)	$PDI_{PPP}-PDI_{PS}$
1	Ph ₂ O	36.6	25	14–24	6.1–7.9	2.3–3.0
2	Ph ₂ O	228	28	73–150	19–27	3.8–5.6
2'	—	—	—	100–220	65–120	1.5–1.8
3	Ph ₂ O	1410	20	220–530	29–41	7.6–13
3'	—	—	—	270–640	160–340	1.7–1.9
4	melt	—	1.5	150–350	29–41	5.2–8.5
4'	—	—	—	150–380	69–120	2.2–3.1
5	melt	—	5.0	250–620	32–45	7.8–14

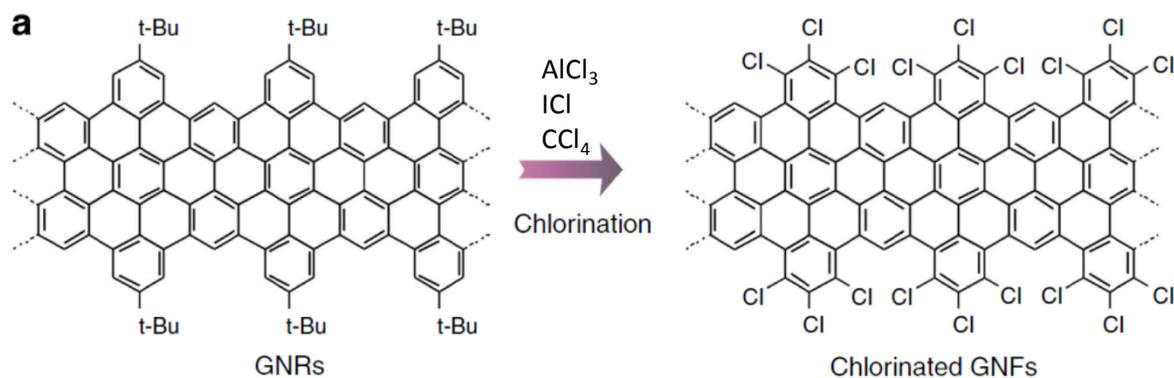
Although purified, M_n is very high



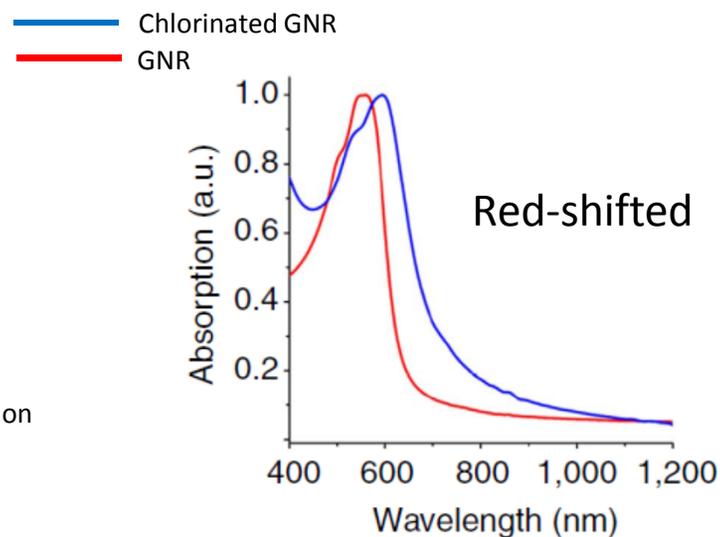
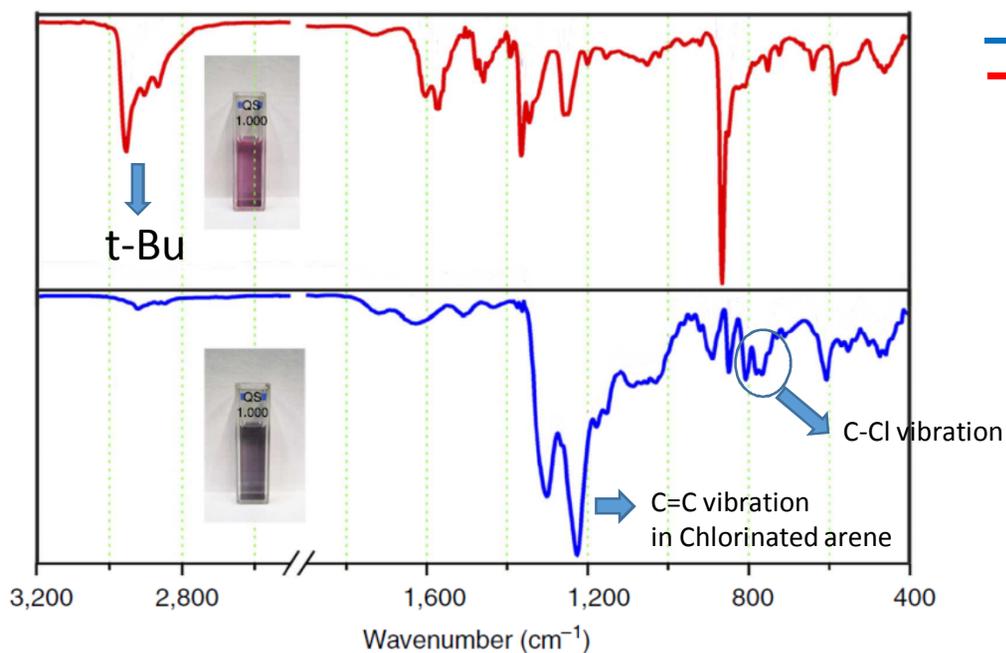
After fractionation by prep GPC

Graphene Nanoribbons (Solution Synthesis)

Chlorination of GNR reported in NChem2014

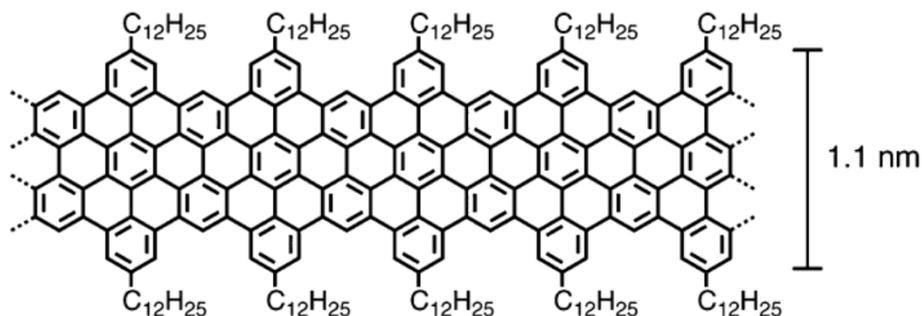


Solubility increases
0.05 mg/mL
in NMR or trichlorobenzene



Graphene Nanoribbons (Solution Synthesis)

Conductivity from GNR of NChem2014



Mobility itself is good
Quantum Yield is low

Figure 1. Chemical structure of the GNRs investigated. Quantum confinement in the lateral dimension induces a bandgap of 1.88 eV.

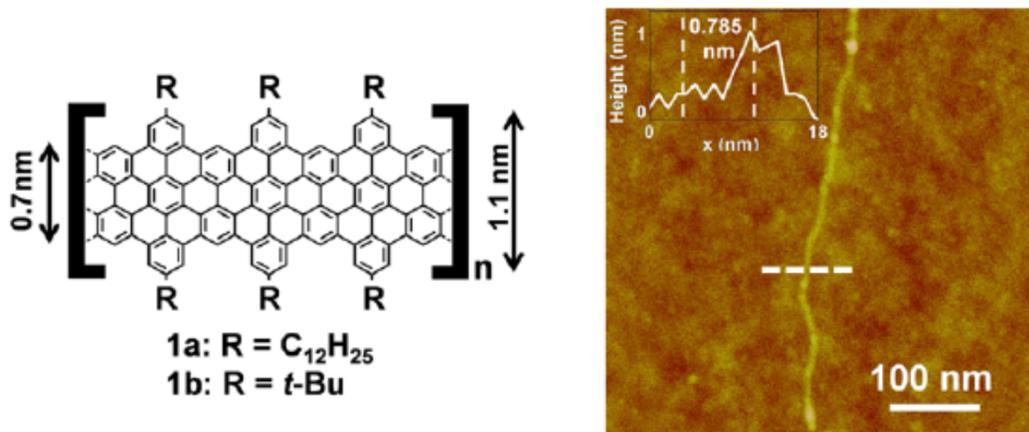
Table 1. Fit Parameters from the Probe Frequency-Dependent GNR Data and CNT Conductivity Fitted to the Drude–Smith Model, eq 1^a

	c	τ (fs)	QY (%)
GNR dispersion	-0.92 ± 0.01	30 ± 3	3 ± 1
GNR film	-0.79 ± 0.07	35 ± 20	4 ± 3
CNTs in gel	-0.90 ± 0.02	170 ± 50	15 ± 10
CNT film	-0.72 ± 0.05	150 ± 15	27 ± 10

Our findings of longer scattering times and higher free carrier generation quantum efficiency in CNTs as compared to GNRs suggest that CNT-based (opto)-electronic devices will likely be more efficient than GNR-based ones.

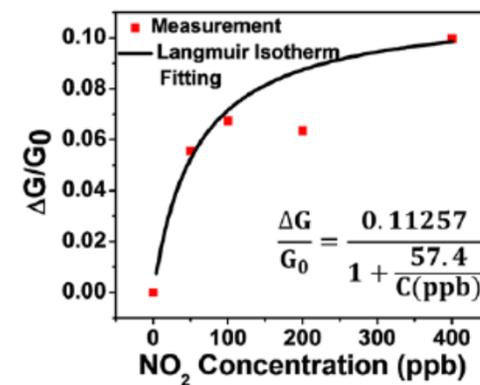
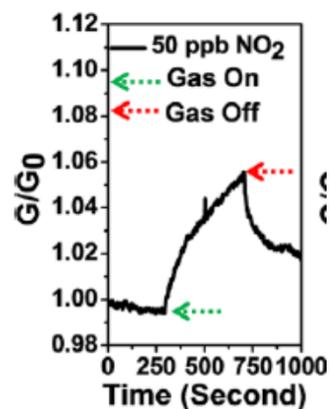
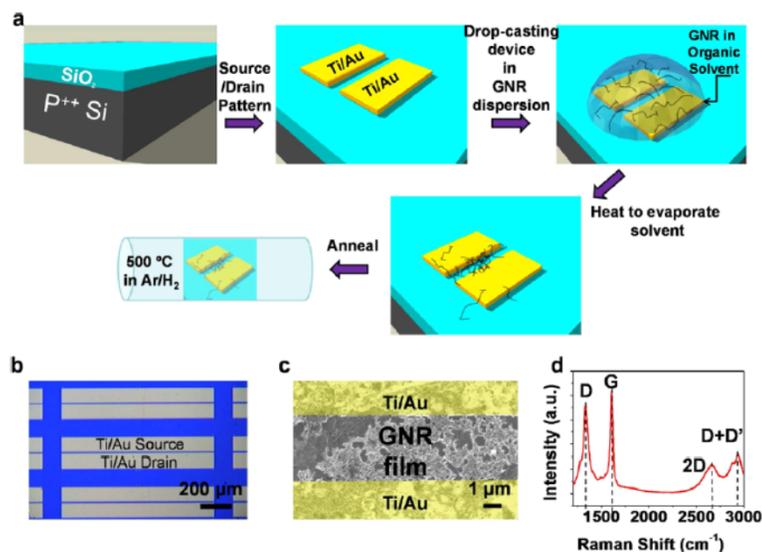
Graphene Nanoribbons (Solution Synthesis)

Single-GNR-based device : NO₂ sensing



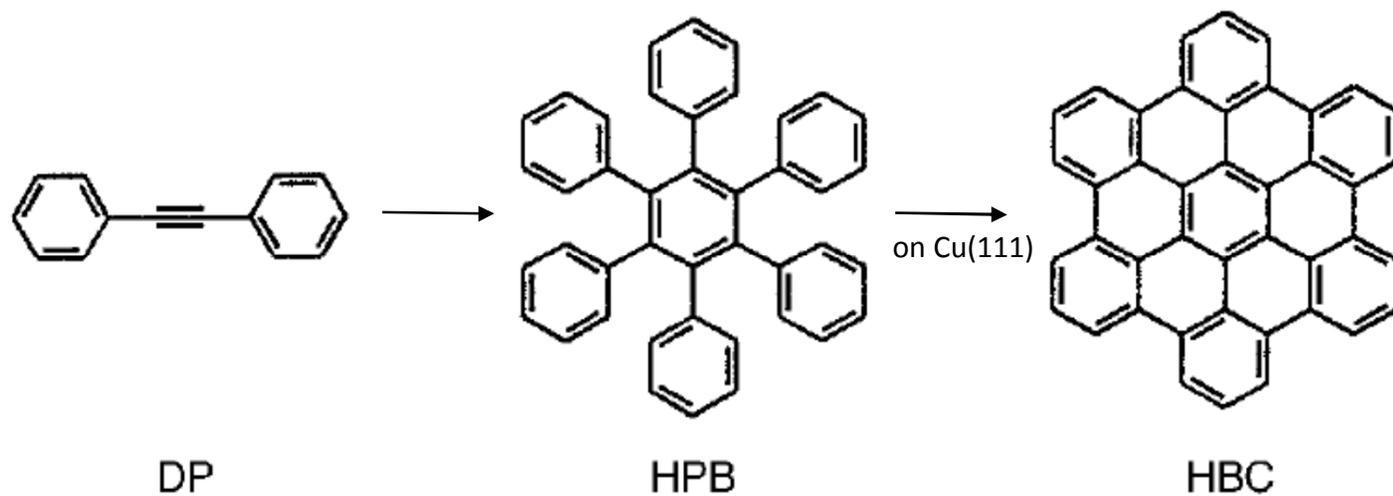
500 nm length

Conductance (G)
increases by electron extraction from NO₂



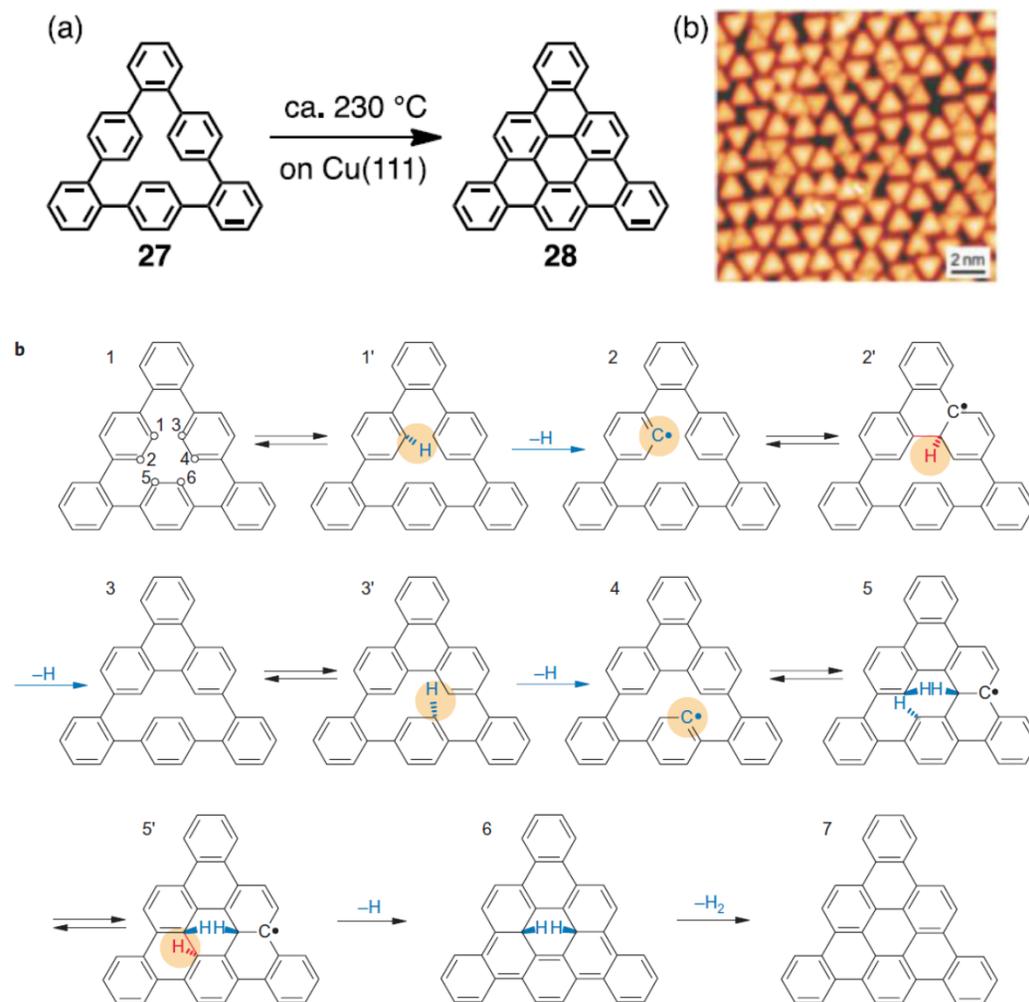
Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis in 1999.



Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis in 2010 with Fasel group.

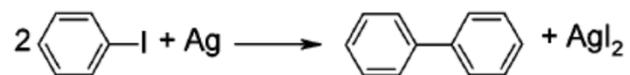


Surface-assisted cyclodehydrogenation

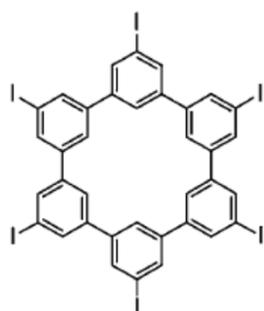
Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis in 2009 with Fasel group.

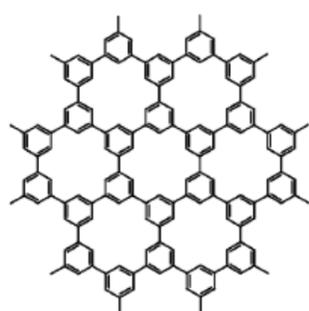
On Ag(111) surface at 200 °C



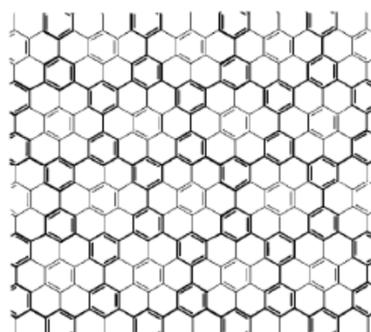
A On-surface coupling of aryl halide



B

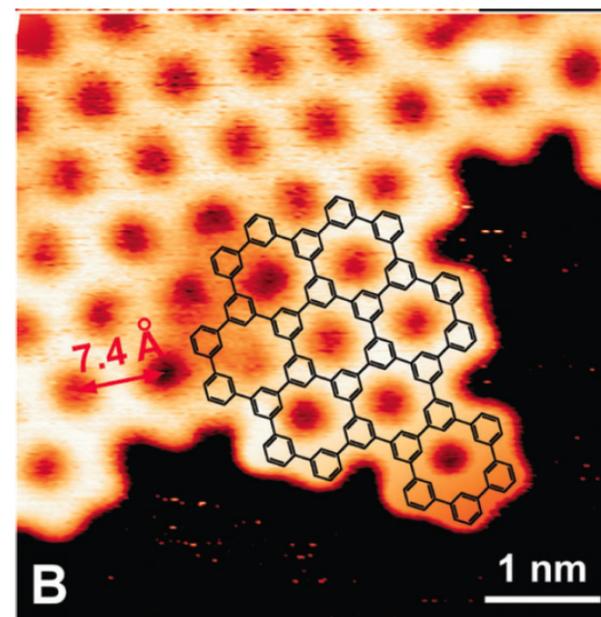


C



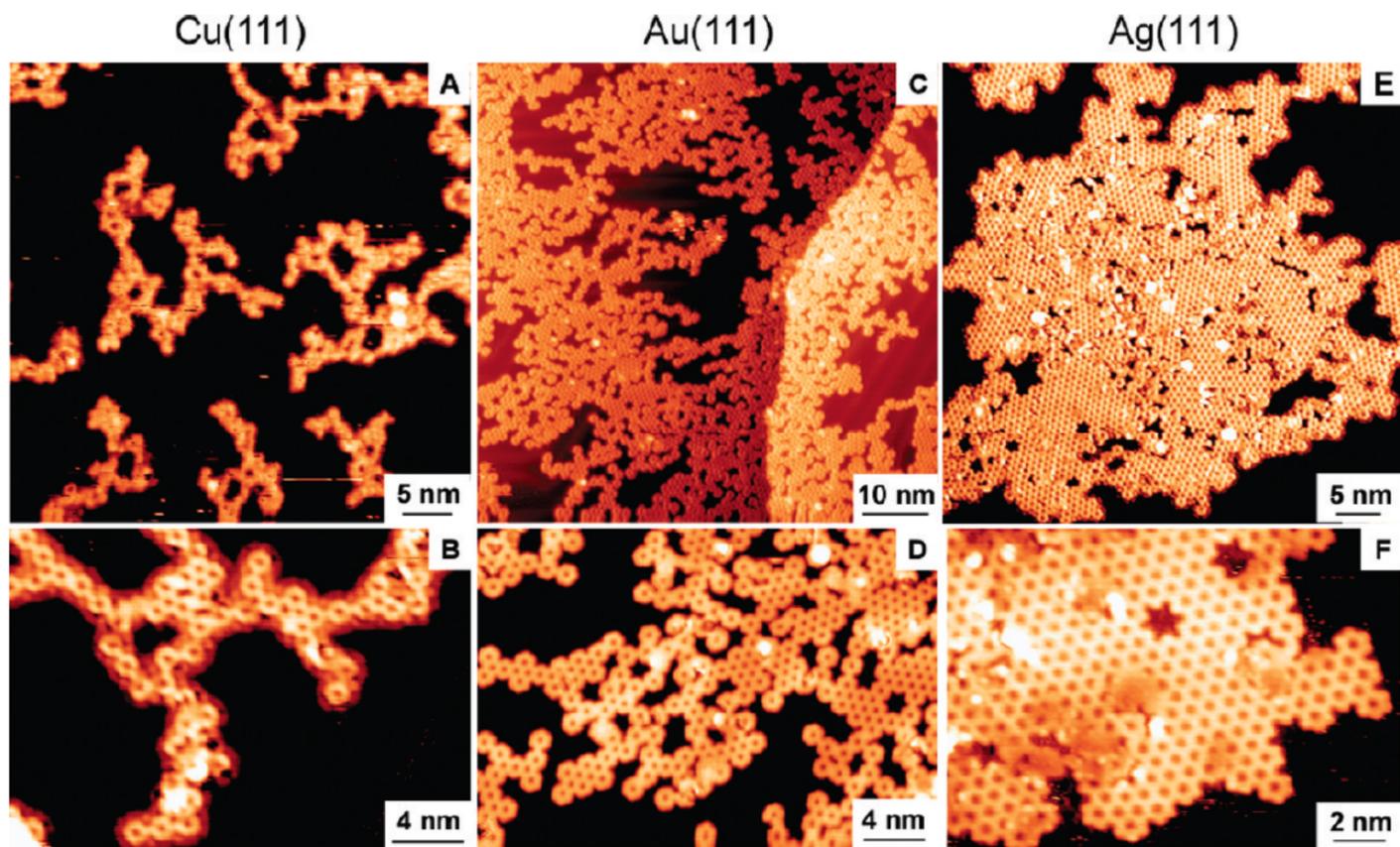
D

Radical formation on surface, and then radical polymerization



Graphene Nanoribbons (Surface Synthesis)

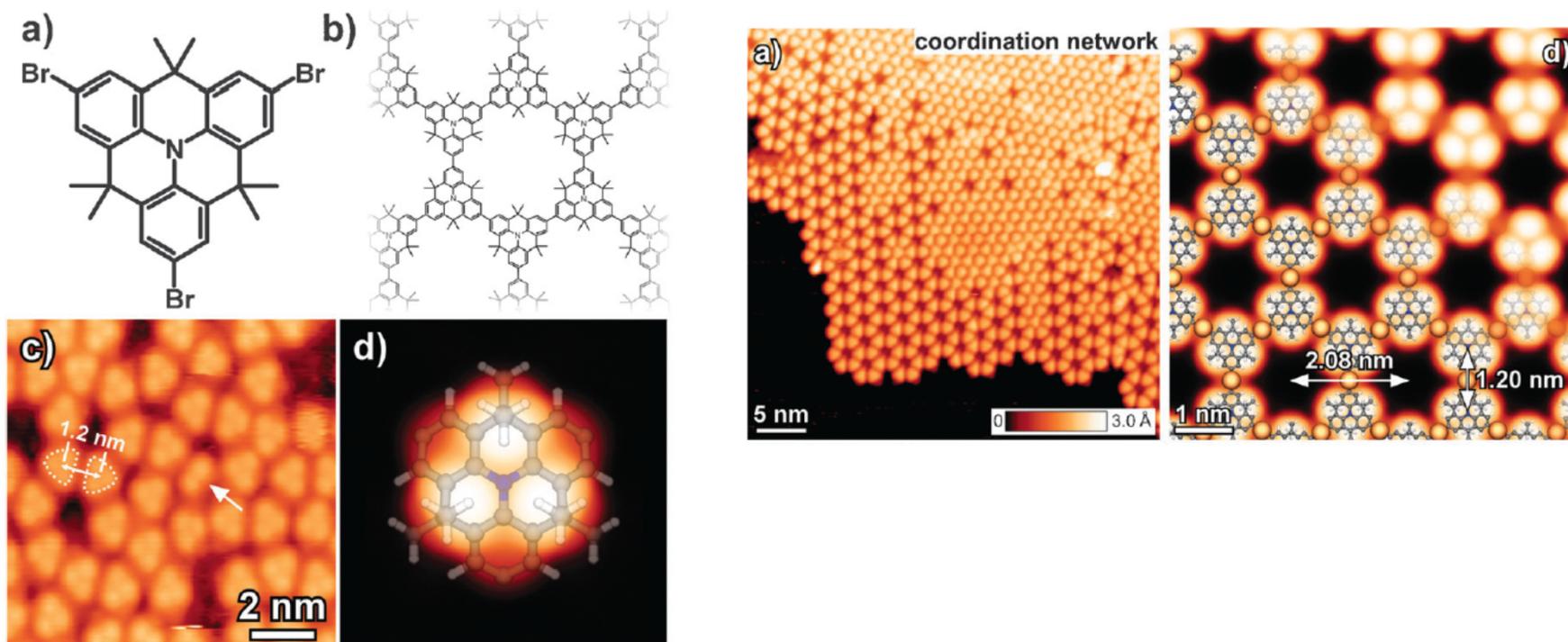
Surface-assisted synthesis in 2010 with Fasel group.



The choice of surface plays a crucial role,
Mainly because of the balance between diffusion and intermolecular coupling

Graphene Nanoribbons (Surface Synthesis)

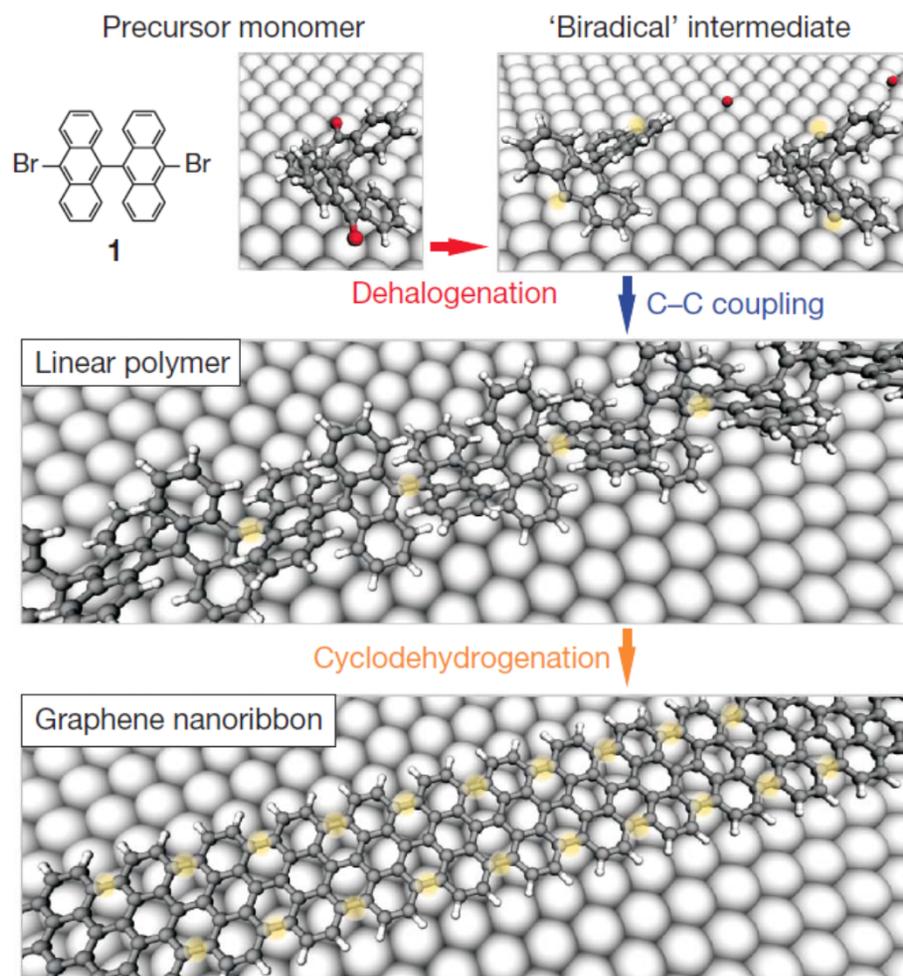
Surface-assisted synthesis in 2010 with Fasel group.



N-doped version

Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis of graphene nanoribbon in 2010.



Typical procedure

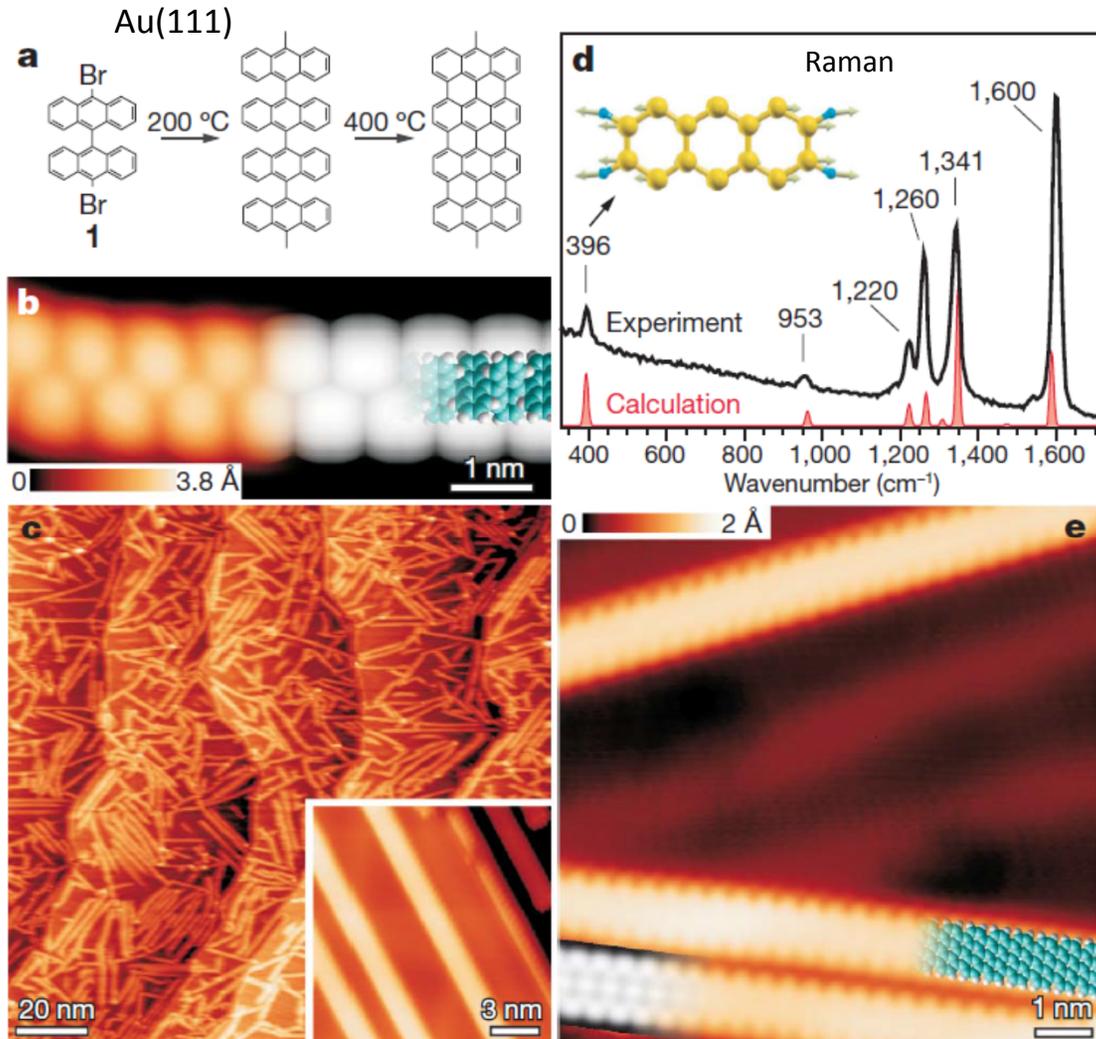
- i) Vacuum sublimation of dihalogenated monomers onto a metal surface
- ii) Biradical monomer formation by thermal cleavage on surface
- iii) Radical polymerization of the surface-stabilized biradical intermediates
- iv) Surface-assisted cyclodehydrogenation by annealing at a higher temperature.

Advantage

- no worries about solubility
- in situ STM monitoring

Graphene Nanoribbons (Surface Synthesis)

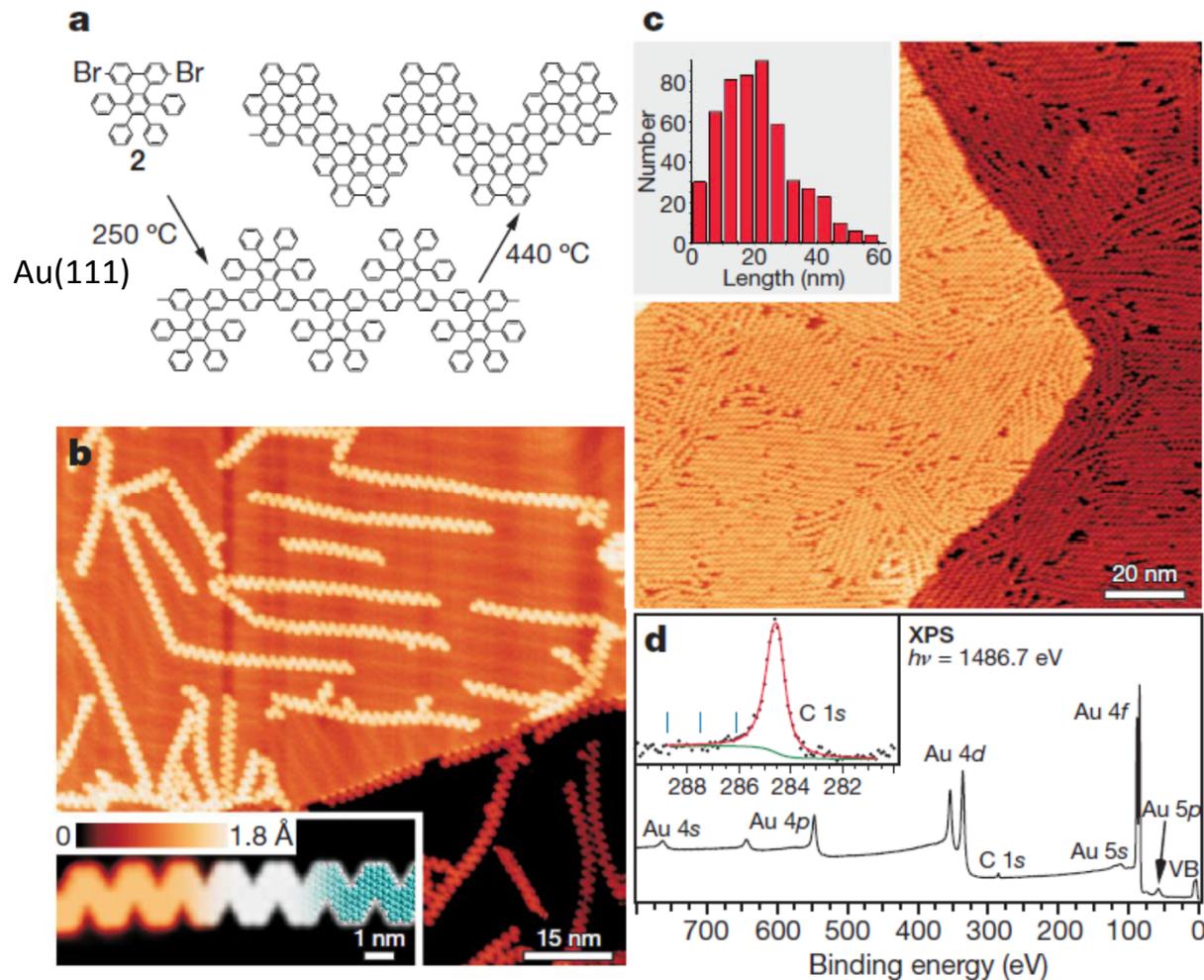
Surface-assisted synthesis of graphene nanoribbon in 2010.



Atomically precious
and very long GNR

Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis of graphene nanoribbon in 2010.

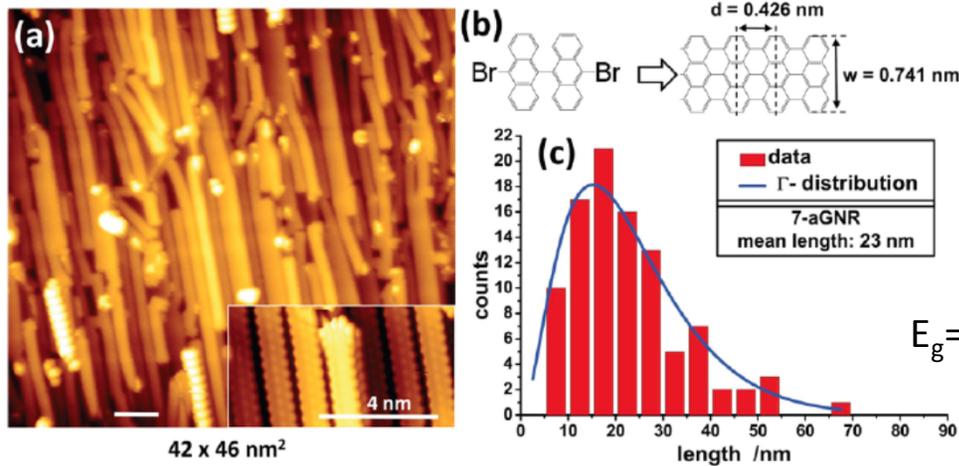


Chevron-type GNR

Chemically pure

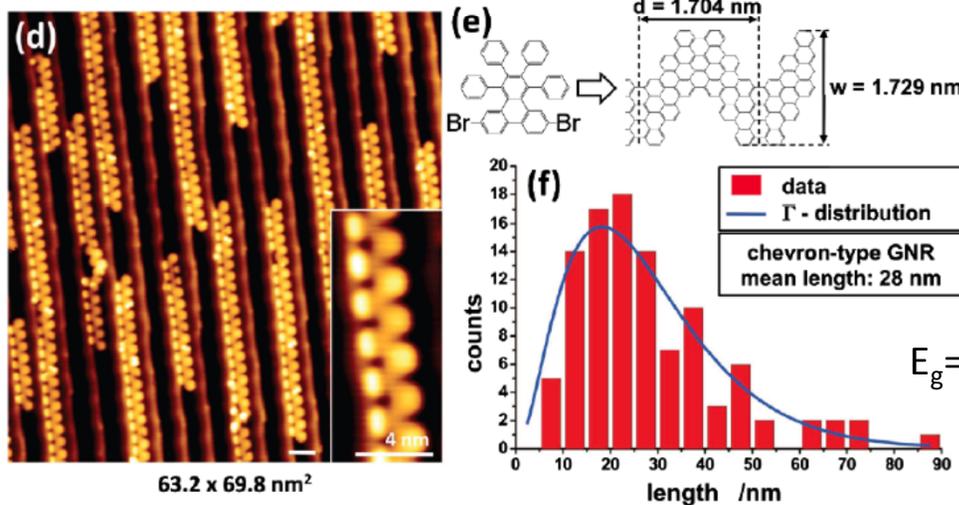
Graphene Nanoribbons (Surface Synthesis)

Surface-assisted synthesis of graphene nanoribbon in 2012.



On Au(788) surface,
Spatially aligned GNRs

$E_g = 2.8$ eV on Au(788),
 2.3 eV on Au(111)

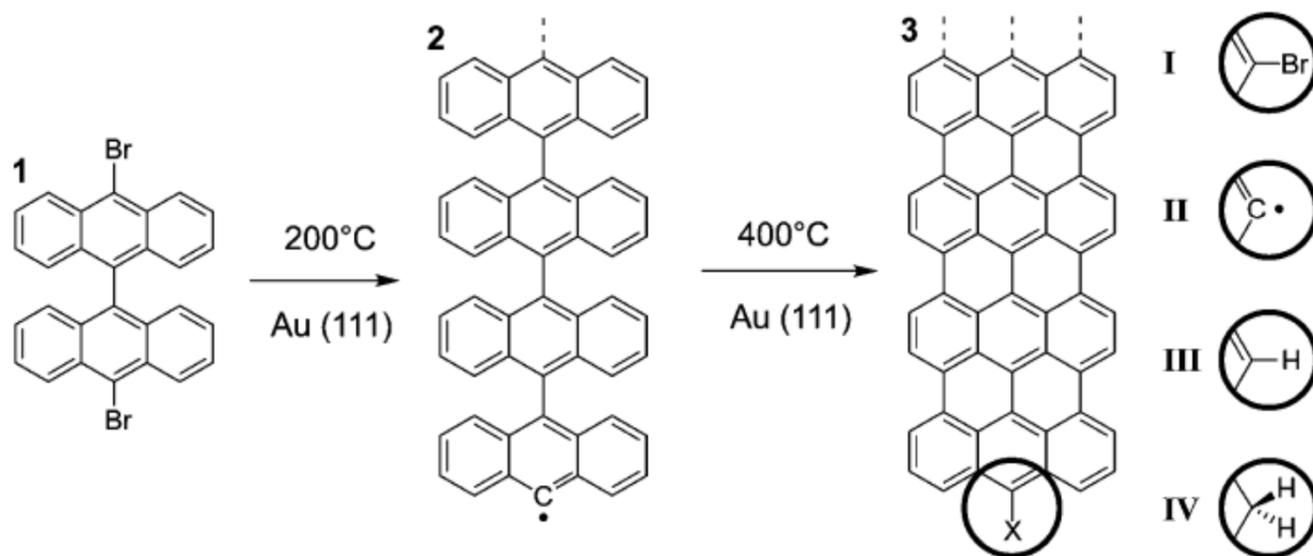


$E_g = 3.1$ eV

Graphene Nanoribbons (Surface Synthesis)

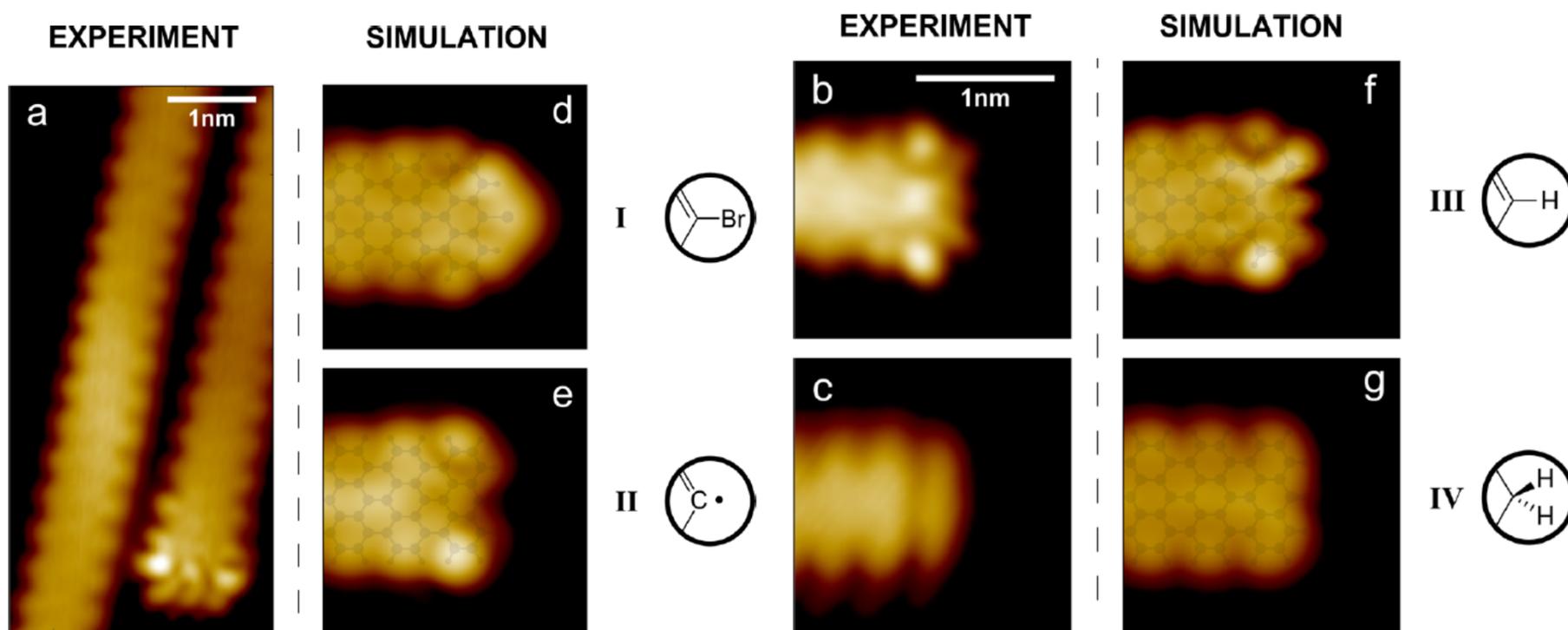
Termination factor of Surface-assisted GNR synthesis in 2013.

Scheme 1. On-Surface Synthesis of 7-AGNRs with Plausible Terminations I–IV



Graphene Nanoribbons (Surface Synthesis)

Termination factor of Surface-assisted GNR synthesis in 2013.

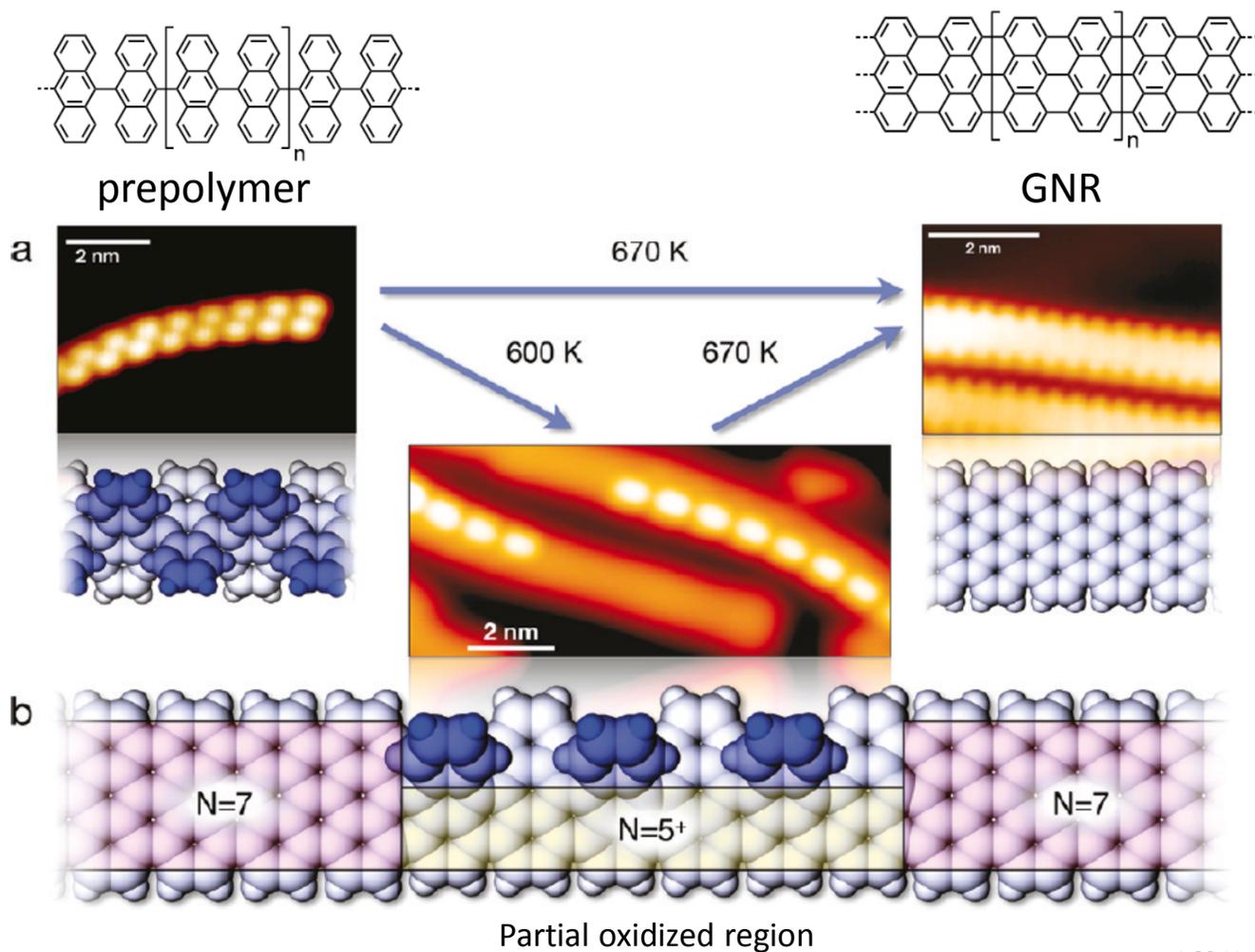


Polymer growth is terminated by the passivation of the radicals with hydrogen

=> Suppressing the spontaneous generation of hydrogen during the radical polymerization step is crucial

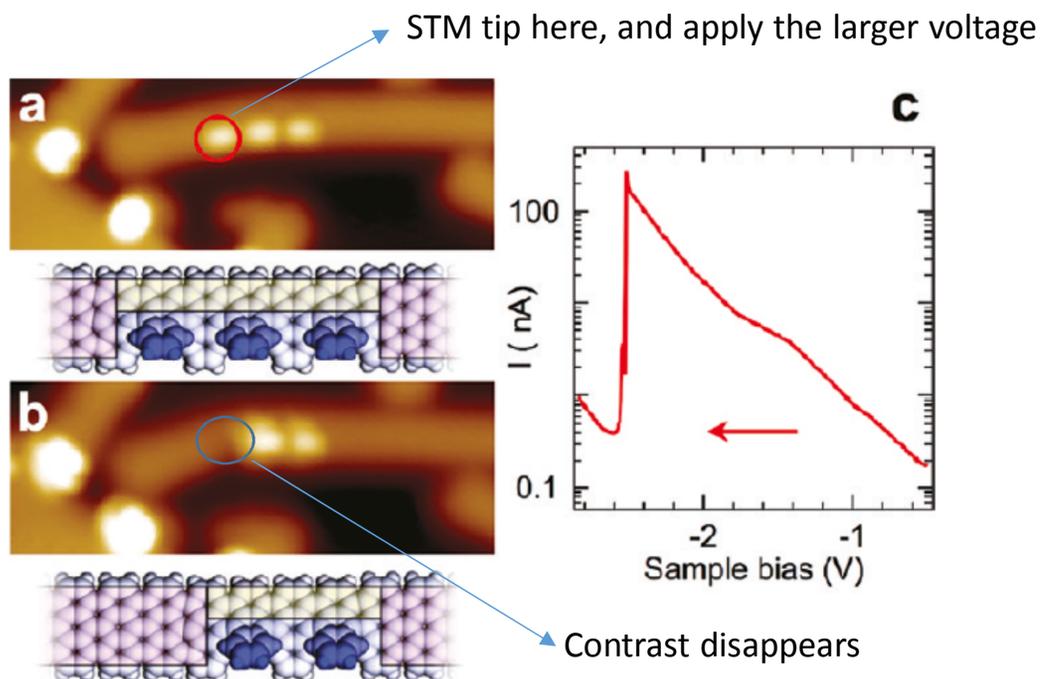
Graphene Nanoribbons (Surface Synthesis)

Intraribbon heterojunction formation in 2012.



Graphene Nanoribbons (Surface Synthesis)

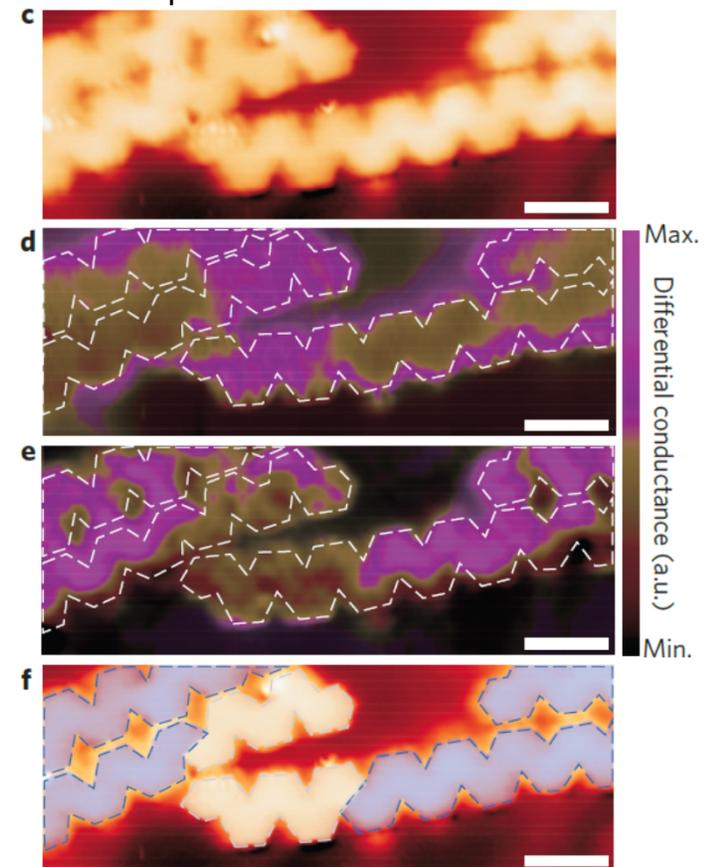
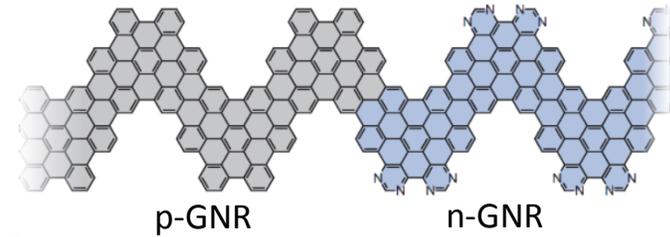
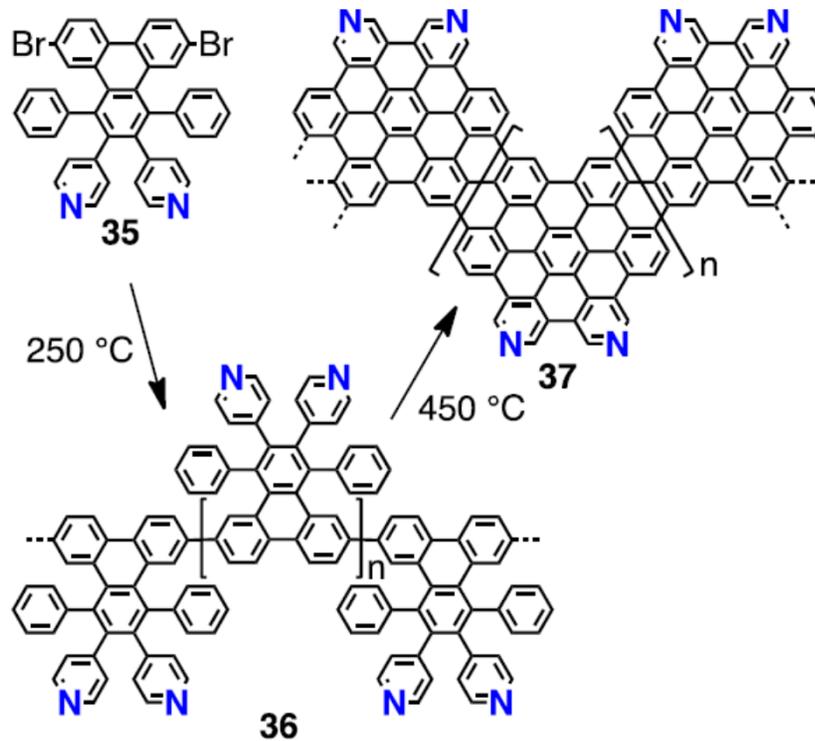
Intraribbon heterojunction formation in 2012.



STM Tip-induced dehydrogenation
(electron-stimulated cyclodehydrogenation)

Graphene Nanoribbons (Surface Synthesis)

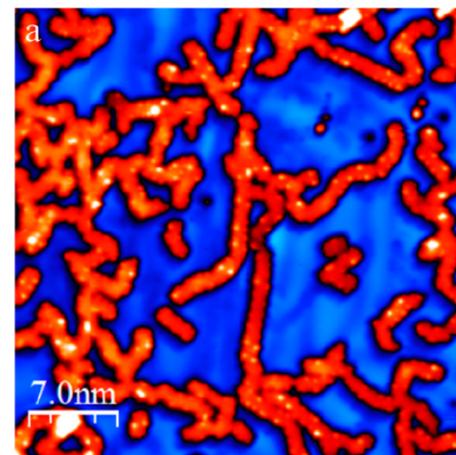
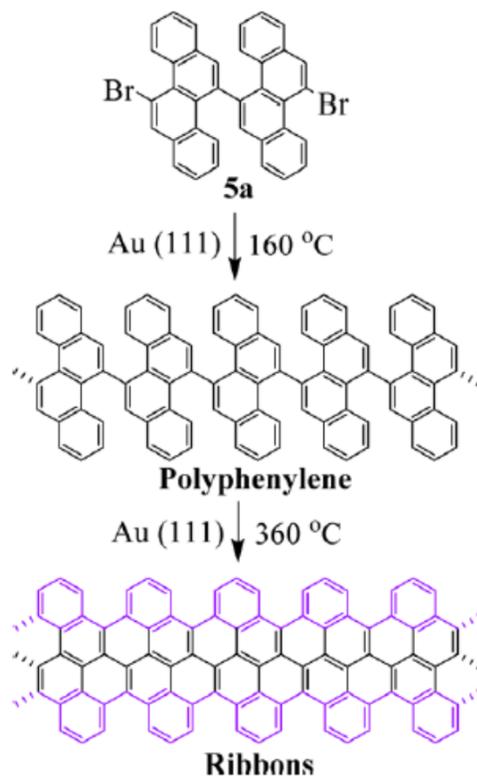
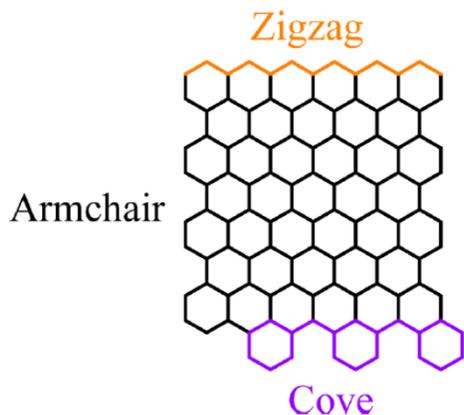
Graphene nanoribbon heterojunction in 2014.



Graphene Nanoribbons (Surface Synthesis)

Cove edge GNR in 2015.

Scheme 1. Edge Structure of Graphene



No empirical data about optical/electrochemical properties

Graphene Nanoribbons (Surface Synthesis)

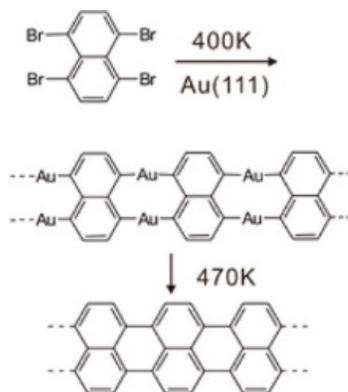
$N=3p+2$ GNR in 2015.

Three subfamilies of GNR

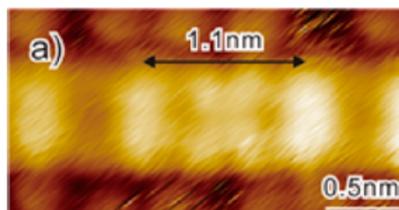
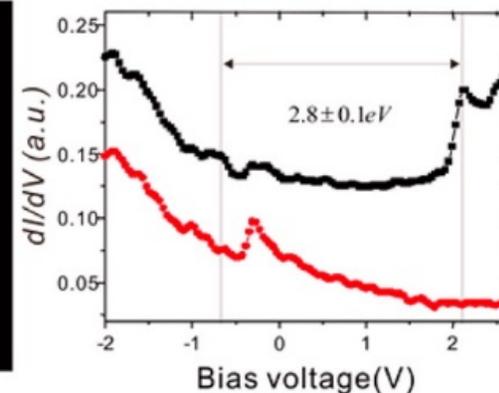
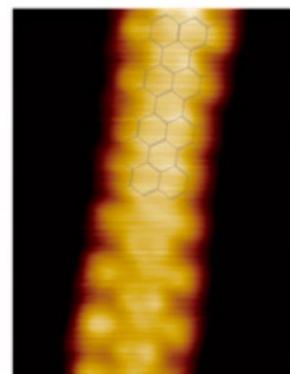
$N=3p$

$N=3p+1$

$N=3p+2 \rightarrow$ rare

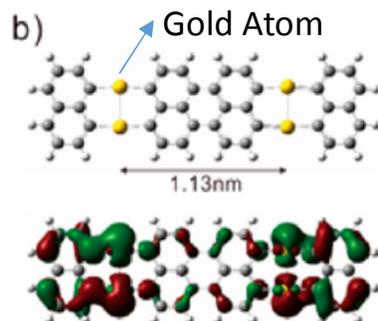


N=5 GNR



Not direct radical coupling,

Au directly involved C-C coupling.



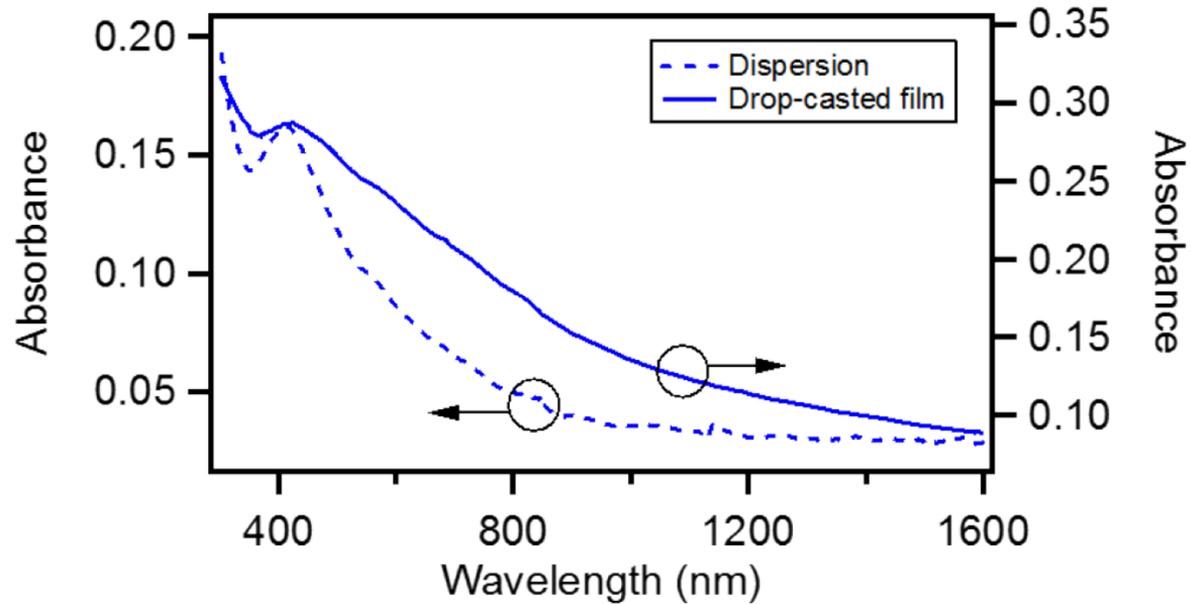
Thank you

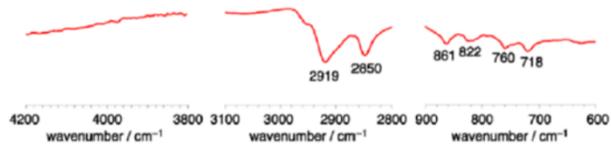
1. UV/vis absorption spectroscopy is one of the nice methods to calculate bandgap between valence band (or HOMO) and conduction band (or LUMO). If we obtain a UV/vis spectrum, we can easily calculate the bandgap of material by using the simple equation, $\text{bandgap (eV)} = 1240/\lambda_{\text{onset}} \text{ (nm)}$.

When the optical bandgap is known as 1.6 eV, what is the expected UV/vis spectrum?

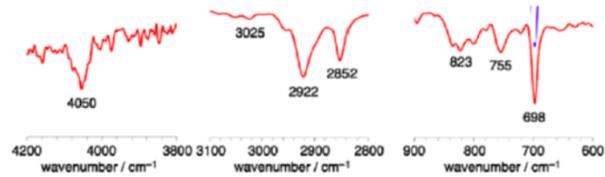
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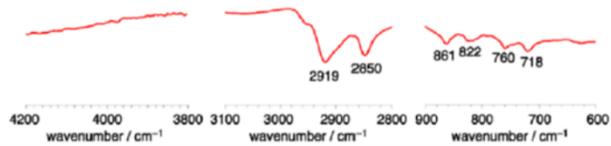
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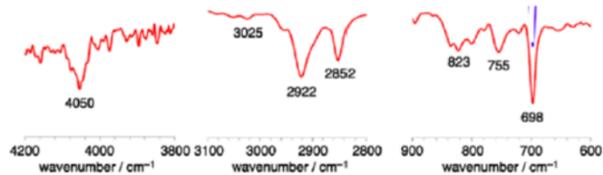


When we make graphene nanoribbons in a bottom-up approach, IR absorption spectroscopy is one of the good tools to determine whether cyclodehydrogenation is successful or not. Which one is the spectrum of polymers before cyclodehydrogenation, and which one is the spectrum of polymers after cyclodehydrogenation? Why?





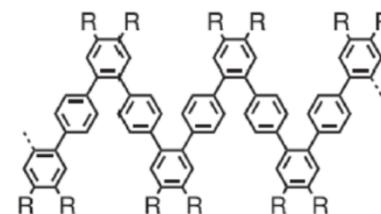
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Upper : after cyclodehydrogenation

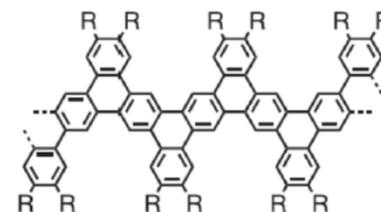
Lower : before cyclodehydrogenation

3. GPC gives us a relative average molecular weight of the polymer compared to the standard polymer (usually polystyrene). Different size of polymer has different retention time on GPC trace, so we can calculate MW using the calibration curve of polystyrene. However, retention time actually depends on not the real size of polymers but the **hydrodynamic size** of polymers (size of solvated polymer including solvent). Usually more flexible polymers tend to have smaller hydrodynamic volume because they are likely to be a random-coil state (like noodles). With these in mind, explain the huge difference of GPC MWs (9900 Da vs. 20000 Da) from the polymers **16b** and **17b**



16a: R = H

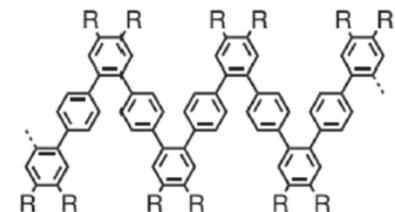
16b: R = C₁₂H₂₅



17a: R = H

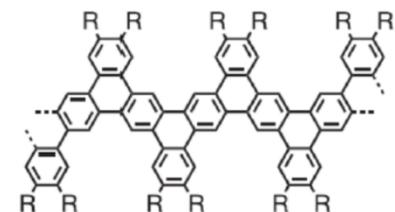
17b: R = C₁₂H₂₅

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16a: R = H

16b: R = C₁₂H₂₅



17a: R = H

17b: R = C₁₂H₂₅

17b is much more rigid than 16b

⇒ Hydrodynamic volume of 17b is much larger than that of 16b

⇒ Retention time of 17b is much shorter than that of 16b

⇒ GPC shows much higher MW for 17b than 16b

